[Article]

doi: 10.3866/PKU.WHXB201406193

www.whxb.pku.edu.cn

单分散PtNi纳米粒子特殊的红外光学性能

周新文^{1,*} 杜娟娟¹ 孙世刚^{2,*}¹三峡大学生物与制药学院, 湖北 宜昌 443002;²厦门大学固体表面物理化学国家重点实验室, 化学化工学院化学系, 福建 厦门 361005)

摘要: 采用电置换反应以及化学还原法制备了单分散PtNi纳米粒子, 循环伏安结果显示该纳米粒子在0.1 mol·L⁻¹硫酸介质中对CO的氧化表现出比本体Pt电极更好的电催化活性. 以CO为探针分子, 采用电化学原位红外光谱研究了PtNi纳米粒子上的特殊红外光学性能. 结果表明, PtNi纳米粒子无论是在玻碳电极还是在金电极上, 均表现出对称的双极谱峰, 同时给出很强的增强效应. 论文研究结果有助于进一步了解低维纳米材料特殊红外性能的本质.

关键词: 电置换反应; 单分散PtNi纳米粒子; 双极红外特征; 原位电化学傅里叶变换红外光谱
中图分类号: O646

Anomalous IR Optical Properties of Monodispersed PtNi Nanoparticles

ZHOU Xin-Wen^{1,*} DU Juan-Juan¹ SUN Shi-Gang^{2,*}¹College of Biological and Pharmaceutical Science, China Three Gorges University, Yichang 443002, Hubei Province, P. R. China;²Department of Chemistry, College of Chemistry and Chemical Engineering, State Key Laboratory of Physical Chemistry of Solid Surface, Xiamen University, Xiamen 361005, Fujian Province, P. R. China)

Abstract: Monodispersed PtNi nanoparticles were synthesized by galvanic displacement reaction and chemical reduction. The monodispersed PtNi nanoparticles demonstrate, by cyclic voltammetry, enhanced electrocatalytic properties for CO oxidation in 0.1 mol·L⁻¹ H₂SO₄ solution compared with bulk Pt electrode. *In situ* electrochemical Fourier transform infrared (FTIR) spectroscopy using CO as the probe molecule was studied. The CO adsorbed on either the PtNi/GC (glassy carbon) electrode or PtNi/Au electrode exhibits characteristics of a symmetric bipolar IR feature with a strong enhancement factor. The results of this paper contribute to the understanding of the special properties and origin of the anomalous IR properties of low-dimensional nanomaterials.

Key Words: Galvanic displacement reaction; Monodispersed PtNi nanoparticle; Bipolar IR feature; *In situ* electrochemical FTIR spectroscopy

1 Introduction

Platinum-nickel (PtNi) alloy nanoparticles have a potential application in high density data storage¹ and are used as anode catalysts and cathode catalysts in fuel cell.² For example, PtNi nanoparticles were frequently used as an anode catalyst for electro-oxidation of methanol in direct methanol fuel cell (DMFC),^{3,4} and

as a cathode catalyst for electro-reduction of oxygen.^{5,6} These electrocatalytic properties are highly dependent on the size, shape, and composition of the PtNi nanoparticles. Geometry of a nanomagnet has great impact on its magnetic properties resulting from the interplay among different types of magnetic energies. Then syntheses of PtNi nanoparticles with controllable mor-

Received: May 12, 2014; Revised: June 19, 2014; Published on Web: June 19, 2014.

*Corresponding authors. ZHOU Xin-Wen, Email: xwzhou@ctgu.edu.cn; Tel: +86-717-6395580. SUN Shi-Gang, Email: sgsun@xmu.edu.cn; Tel: +86-592-2180181.

The project was supported by the National Natural Science Foundation of China (21229301, 21403126) and Research Foundation of Education Bureau of Hubei Province, China (D20131302).

国家自然科学基金(21229301, 21403126)和湖北省教育厅重点项目(D20131302)资助

© Editorial office of *Acta Physico-Chimica Sinica*

phology and composition are crucial for progress in this field of data storage, fuel cell and so on.

Until now, rod-shaped superparamagnetic PtNi nanoalloy and PtNi/C nanoparticles have been synthesized through different methods.^{7,8} Weller and coworkers⁹ have reported the synthesis of $\text{Ni}_x\text{Pt}_{1-x}$ nanoparticles with tunable composition and size by a hot organometallic synthesis, but the method needs a high temperature and expensive poisonous reactants. PtNi hollow spheres prepared by a template-replacement route as catalysts for hydrogen generation from ammonia borane have been investigated by Chen *et al.*¹⁰ This method consists of the removal of the template after the synthesis, which may destroy the structure of the products. Except for the above synthetical methods, galvanic displacement reaction is a special template method, in which one substance is served as a suitable sacrificial template and reacts with other appropriate metal ions according to their different standard reduction potentials, resulting in the controlled formation of nanomaterials.^{11–13} Bai and coworkers¹² have reported the synthesis of Pt hollow nanospheres using Co nanoparticles as a sacrificial template. Co@Pt nanoalloy,¹⁴ hollow superparamagnetic CoPt nanospheres,¹⁵ and necklace-like noble-metal hollow nanoparticle chains (Au, Pd, Pt)¹⁶ were also prepared using the similar method. The galvanic displacement reaction has also been used for the synthesis of PtNi nanoparticles with a hollow structure.^{17,18}

The adsorbed molecules on these nanomaterials' surfaces yield anomalous IR features named the abnormal IR effects (AIRES),¹⁹ enhanced IR absorption (EIRA),²⁰ and the Fano-like IR effects.^{21,22} Further investigation has showed that nanostructured Ni, Co thin film and CoPt nanorods also display anomalous IR properties.^{23–25} It is important to note that the study of anomalous IR properties of nanomaterials with different morphologies and composition is of great importance in revealing the fundamentals of nanomaterials.²⁶

In this paper, we report the synthesis of PtNi nanoparticles by galvanic displacement reaction. Electrocatalytic and IR optical properties are studied in detail. The results of *in situ* IR spectra show that the adsorbed CO on monodispersed PtNi nanoparticles exhibit characteristics of bipolar IR feature with a strong enhancement.

2 Experimental

2.1 Preparation of PtNi nanoparticles

All chemicals used in this experiment were analytical grade and were used without further purification. The growth of the PtNi nanoparticles was carried out in a solution-phase system reported in the literature.¹⁷ In brief, 9.5 mg NiCl_2 , 60 mL H_2O , and 100 mg poly(vinylpyrrolidone) (PVP) were mixed in a three-necked flask equipped with a heat controller. The mixture was kept at 75 °C with vigorous stirring. Then, 25 mL of 0.01 mol $\cdot$ L⁻¹ freshly prepared NaBH_4 solution was added dropwise. The solution turned dark, indicating the formation of Ni. Once the NaBH_4 is completely dropped into the solution, 20 mL of 4 mmol $\cdot$ L⁻¹ K_2PtCl_6 was added dropwise. To avoid oxidation of the Ni nanoparticles,

high-purity N_2 was bubbled into the solution during the whole procedure. The obtained black solution was stirred for 2 h at the constant temperature of 75 °C and cooled to room temperature. The obtained PtNi nanoparticles were separated, washed thoroughly, and redispersed in Millipore H_2O for following studies.

Transmission electron microscopy (TEM) images were obtained on instruments of JEM-100CX-II electron microscopy.

2.2 Electrochemical *in situ* FTIR spectroscopy

Glassy carbon (GC) electrode, polycrystalline Pt (bulk Pt) electrode, and polycrystalline Au (bulk Au) electrode were polished mechanically using sandpaper and alumina powders of sizes 5 and 1 μm down to 0.3 μm to obtain a clean electrode surface. A prescribed quantity of suspension of PtNi nanomaterials was applied to the surfaces of GC and Au electrodes, upon which a drop of 0.5% (*V/V*) Nafion solution was dispersed to fix the nanomaterials on the surface. The electrodes thus prepared are denoted as PtNi/GC and PtNi/Au electrodes, respectively. The cyclic voltammetric (CV) studies were carried out with a CHI631C electrochemical work station (CH instruments, Inc.), at a potential scan rate 50 mV $\cdot$ s⁻¹. Electrochemical experiments were carried out in a standard three-electrode cell at room temperature. The counter electrode was a Pt flake and reference electrode was a saturated calomel electrode (SCE).

The IR optical properties of the PtNi nanomaterials were studied and CO was selected as a probe molecule. The multistep FTIR (MS-FTIR) procedure²⁷ was used in the present study. In brief, a series of single-beam spectra were collected first at sample potentials (E_s), where adsorbed CO is stable, and a single-beam spectrum was collected finally at the reference potential (E_r), at which adsorbed CO has been removed completely by electro-oxidation. In our *in situ* FTIR measurements, the reference potential (E_r) was settled as 1.2 V, the sample potentials were settled as -0.2, -0.1, 0.0, 0.1, 0.2, and 0.3 V based on the results of CO oxidation on PtNi/GC electrode. The resulting spectra were calculated using the following equation:

$$\frac{\Delta R}{R} = \frac{R(E_s) - R(E_r)}{R(E_r)} \quad (1)$$

where $R(E_s)$ and $R(E_r)$ are single-beam spectra of reflection collected at sample potential E_s and reference potential E_r , respectively. Each single-beam spectrum was recorded by collecting and co-adding 400 interferograms at a spectral resolution of 8 cm⁻¹. *In situ* FTIR spectroscopic measurements were carried out on a Nexus 870 FTIR spectrometer (Nicolet) equipped with an Ever-Glo IR source and a liquid nitrogen-cooled MCT-A detector. A CaF_2 disk was used as the IR window, and a thin layer of solution was formed by moving the electrode toward the CaF_2 window during FTIR measurements.

3 Results and discussion

TEM images in Fig.1 illustrate that large-scale monodispersed PtNi nanomaterials have been synthesized. The structural details are revealed in high-magnification TEM images. The PtNi nanomaterials are solid, spherical, and highly monodispersed with

Fig.1 TEM images and the size distribution of PtNi nanoparticles

The insets show the enlarged TEM image and the size distribution.

an average diameter of about 52.2 nm.

Fig.2 shows the cyclic voltammograms of adsorbed CO oxidation on the PtNi/GC electrode in 0.1 mol·L⁻¹ H₂SO₄ solution. It can be seen that the hydrogen adsorption-desorption current is completely suppressed when the electrode surface is covered with CO. The onset potential of CO oxidation is measured at about 0.12 V, and the main oxidation current peak appears around 0.51 V. It is worthwhile to note that, under the same conditions, the onset potential of CO oxidation on PtNi/GC electrode has been shifted negatively by 300 mV in comparison with that on bulk Pt, and the peak potential (E_p) on PtNi/GC electrode has been shifted negatively by 75 mV referring to the value measured on bulk Pt. In the reverse scan, a reduction current peak of Pt oxide is recorded at 0.36 V. The results demonstrated that the PtNi nanomaterials exhibit better electrocatalytic properties for the CO oxidation in 0.1 mol·L⁻¹ H₂SO₄ solution than bulk Pt does.

Fig.3 shows the MS-FTIR spectrum of CO adsorbed on the PtNi/GC and bulk Pt electrodes in 0.1 mol·L⁻¹ H₂SO₄ solution. The negative-going band around 2072 cm⁻¹ is assigned to IR absorption of linearly bonded CO (CO_L). Except for the band of CO_L, a positive-going band located near 2341 cm⁻¹ is ascribed to IR absorption of CO₂ species that are uniquely derived from the

Fig.2 Cyclic voltammograms of adsorbed CO oxidation on the PtNi/GC (solid line) and bulk Pt (dash line) electrodes in 0.1 mol·L⁻¹ H₂SO₄ solution with a scan rate of 50 mV·s⁻¹

Fig.3 *In situ* FTIR spectra of CO adsorbed on NiPt/GC electrode in 0.1 mol·L⁻¹ H₂SO₄ solution

$E_R=1.2$ V, E_s are indicated for each spectrum.

oxidation of adsorbed CO₂ (CO_{ad}) at E_R , because no CO₂ and CO species presented initially in solution inside the thin layer between the electrode and IR window. It is necessary to note that the IR band of CO_{ad} on a bulk Pt electrode always appears in the opposite direction to the CO₂ band under the same conditions, which is in accordance with the predication of Eq.(1), which shows that the adsorbed CO on bulk Pt electrode exhibits a normal IR adsorption.²⁵ From the results of Fig.3, we can see that the IR features of PtNi nanoparticles are different from those of CO_{ad} on the bulk Pt electrode and shown a typical IR features of AIREs,^{19,25} which have three distinct anomalous features: (1) the inversion of the direction of adsorbed CO; (2) the significant enhancement of IR absorption; and (3) the broadening of the width of adsorbed CO bands.

In the *in situ* FTIR spectroscopic measurements, we have emphasized that the adsorbed CO will be removed completely by electrooxidation at reference potential ($E_R=1.2$ V). Fig.4 shows the *in situ* single-beam spectra of reflection collected at reference potential. In Fig.4 (line a), a bipolar band near 2100 cm⁻¹ is assigned to IR adsorption of CO_L, which means that the adsorbed CO was not electrooxidized completely at reference potential. So the results obtained from Fig.3, which were calculated from Eq. (1) using the line a in Fig.4 as the $R(E_R)$, is questionable. Then we continue collecting the single-beam spectra of reflection at E_R . At

Fig.4 *In situ* single-beam spectra of reflection collected at E_R

The line a is the reference spectrum $R(E_R)$; The line b is another reference spectrum $R(E_{R,2})$.

last, the bipolar band near 2100 cm^{-1} was disappeared indicating that the adsorbed CO was electrooxidized completely (line b). Then we can obtain the final resulting spectra using the line b in Fig.4 as the $R(E_{R-2})$ from Eq.(1) based on the above analyses. Fig.5 shows the final *in situ* FTIR spectra of CO adsorbed on NiPt/GC electrode in $0.1\text{ mol}\cdot\text{L}^{-1}\text{ H}_2\text{SO}_4$ solution. The E_R is 1.2 V, E_S are indicated for each spectrum. The dash line is the *in situ* FTIR spectra of CO adsorbed on bulk Pt electrode obtained under the same condition ($E_R=1.2\text{ V}$, $E_S=0.0\text{ V}$). For the PtNi/GC electrode, remarkably enhanced symmetric bipolar band with its positive peak near 2050 cm^{-1} and negative band near 2100 cm^{-1} is assigned to IR adsorption of CO_L adsorbates at E_S that is varied from -0.2 to 0.3 V . It is worthwhile to point out that the bipolar band observed in the current studies shows opposite direction in the positive and negative peaks in comparison to our previous studies of nanostructured Co thin films²⁴ and shows uniform direction to our previous studies of nanostructured Pt thin films.²⁶ The obtained results demonstrate that the bipolar IR feature of the CO_{ad} bands is a particular IR property of nanostructured materials.

The obtained IR feature is obviously different from the normal absorption on the bulk Pt electrode (Fig.5, dash line). The intensity, directions, and the broadening of the width of CO_L are different. This kind highly symmetric bipolar IR feature is also different from the Fano-like line shape, which is asymmetric and has been observed in transmission IR spectra of CO adsorbed on smooth iron ultrathin films supported on Mg(001) under ultra-high-vacuum conditions by Pucci and co-workers.²⁸ The Fano-like line shape has also been observed in attenuated total reflectance (ATR) spectra of CO adsorbed on ultrathin film of Pt substrated on silicon under electrochemical conditions²⁹ and nanostructured Ni,²³ Co,²⁴ Pt,²⁶ and Ru film.³⁰ This kind Fano-like asymmetric spectrum was ascribed to the Fano effects,²¹ and interpreted as the non-adiabatic interaction of adsorbate vibrations with electronic excitations by Pucci and co-workers.²⁸

In Fig.5, except for the adsorbed band of CO_L , a positive-going band located near 2341 cm^{-1} is ascribed to IR absorption of CO_2 species. The intensity of CO_2 species adsorbed on PtNi/GC

electrode is much more weak than that of bulk Pt electrode. The reasons can be explained as follows. It has been testified that the CO_2 species derived from CO_{ad} oxidation can remain in the thin layer solution between the electrode and IR window for a relatively longer time due to large transporting resistance of the dispersion of CO_2 between the bulk solution and the thin layer.³¹ In our case, the time remains at E_R on the PtNi/GC electrode is long enough to allow CO_2 to diffuse from the thin layer to the bulk solution. As a result, the intensity of CO_2 decreased, which can be seen from the different intensities of IR absorption band of CO_2 in Fig.5. Except for the band of CO_L and CO_2 , the absorption band around $1800\text{--}1900\text{ cm}^{-1}$ in Fig.5 is the IR absorption of the existence of a relatively high level of gas water molecules in the beam path due to a poor purging of the spectrometer.

Except for the directions of CO_L , we also noted that the intensity of the IR absorption of CO adsorbed on PtNi/GC electrode is significantly enhanced. It is convenient to define an enhancement factor to systematically study this IR property. The quantitative parameter of enhancement factor (Δ_{IR}) is calculated using the following equation:²⁶

$$\Delta_{IR} = \frac{1}{R_f} \frac{A_{\text{CO(PtNi)}}}{A_{\text{CO(Pt)}}} \quad (2)$$

Where $A_{\text{CO(Pt)}}$ signifies the integrated intensity of CO_L band measured in the spectrum of bulk Pt electrode, i.e., the spectrum (dash line) in Fig.5. $A_{\text{CO(PtNi)}}$ is the integrated intensity of CO_L band measured in the spectrum of PtNi/GC electrode. The surface relative roughness (R_f) measured in cyclic voltammetric studies is introduced in Eq.(2) to calibrate the difference of real surface area of PtNi nanoparticles. The enhancement factor Δ_{IR} has been measured on PtNi/GC electrode to be 73.6. The value of the enhancement factor is much greater than others we obtained previously.^{19,20,23–26} It is important to point out that the calculation of enhancement factor Δ_{IR} is glancing because it is hard to define the baseline of the bipolar IR feature of the CO_{ad} bands exactly. But the adsorbed CO shows a strong enhancement, which will be approved clearly from the next *in situ* single beam spectra.

Because the intensity of the IR absorption of CO adsorbed on PtNi/GC electrode is strong enough, we even can observe the optical properties in the *in situ* single beam spectra. Fig.6 shows the *in situ* single beam spectra of CO adsorbed on NiPt/GC electrode in $0.1\text{ mol}\cdot\text{L}^{-1}\text{ H}_2\text{SO}_4$ solution at E_S that is varied from -0.2 to 0.3 V . It can observe clearly that bipolar band with its positive peak near 2050 cm^{-1} and negative band near 2100 cm^{-1} is assigned to IR adsorption of CO_L adsorbates. In our previous work,^{13,19,23–27} it is impossible to observe the peaks of adsorbed CO in the *in situ* single beam spectra because of their weak enhancement. Then we have to obtain the resulting spectral equation (1) in the former studies. In Fig.6, we observe clearly that either the positive peak or the negative peak of the bipolar CO_{ad} bands shifts positively and linearly with increasing E_S due to Stark effect,^{19,26} which indicated that the CO_L was the adsorbed CO molecule, not the CO molecule dissolved in the solution. The Stark tuning rate can be then determined from the plot of the variation of wavenumbers with E_S in the *in situ* single beam

Fig.5 Final *in situ* FTIR spectra of CO adsorbed on NiPt/GC electrode in $0.1\text{ mol}\cdot\text{L}^{-1}\text{ H}_2\text{SO}_4$ solution

$E_R=1.2\text{ V}$, E_S values are indicated for each spectrum. The dash line is the *in situ* FTIR spectrum of CO adsorbed on bulk Pt electrode obtained under the same condition ($E_R=1.2\text{ V}$, $E_S=0.0\text{ V}$)

Fig.6 *In situ* single beam spectra of CO adsorbed on NiPt/GC electrode in 0.1 mol·L⁻¹ H₂SO₄ solution

spectra in Fig.6. The results are shown in Fig.7. It is interesting to notice that the Stark tuning rate of the positive peak and that of the negative peak of the bipolar CO_L bands are not equal. The values of the Stark tuning rate were measured as 6.6 and 10.2 cm⁻¹·V⁻¹ for the CO_L bipolar bands, respectively. The value of the Stark tuning rate is closely correlated with the nanostructured materials,^{24,26} the concentration of electrolyte,³² and solution molecule.³³ When the E_s is above 0.2 V, the variation of wavenumbers with E_s clearly deviated from the linear relationship and decreased slightly. This is obviously due to a partial oxidation of CO adsorbed on the PtNi nanoparticles, as also can be elucidated in CV curves (Fig.2).

It is known that the reflectivity of substrate may influence IR optical features.³⁴ Glassy carbon is a moderately reflecting sub-

Fig.8 *In situ* single beam spectra of CO adsorbed on NiPt/Au electrode in 0.1 mol·L⁻¹ H₂SO₄ solution

strate, while bulk Au electrode belongs to a high reflecting one. We also studied the IR properties of the PtNi/Au electrode under the same conditions as above. It can be seen that the IR bands of CO_L and CO₂ in Fig.8 are appeared in the same direction as those seen in Fig.6, indicating clearly that CO adsorbed on the PtNi/Au electrode also exhibits bipolar IR feature. The values of the Stark tuning rate were measured as 20.1 and 21.8 cm⁻¹·V⁻¹ for the CO_L bipolar bands, respectively (Fig.9). The results demonstrate that the bipolar IR feature of CO_{ad} is a particular IR property of PtNi nanoparticles, and that the influence of the substrate materials on the IR spectral features may be neglected in the present study.

A bipolar band of linearly chemisorbed CO with anomalous shape was reported in 1994 in potential-difference FTIR spectra

Fig.7 Plots of CO_L band centre *versus* E for CO adsorbed on NiPt/GC electrode obtained from the single beam spectra
(a) up-peak; (b) down-peak

Fig.9 Plots of CO_L band centre *versus* E for CO adsorbed on NiPt/Au electrode obtained from the single beam spectra
(a) up-peak, (b) down-peak

of a basal plane graphite disc covered with Pt particles in a $1.0 \text{ mol} \cdot \text{L}^{-1} \text{H}_2\text{SO}_4 + 1.0 \text{ mol} \cdot \text{L}^{-1}$ aqueous methanol solution.³⁵ They explained the anomalous bipolar CO band by a potential-induced migration of chemisorbed CO molecules from terraces at the reference potential to edge or kink sites at the sample potentials. It was assumed that the CO stretching is higher for terraces than for edge or kink sites. Later, Ortiz *et al.*³⁶ thought the explanation of the anomalous bipolar bands is simple and should be interpreted by the laws of reflectance, which means that the moderately reflecting substrate (GC) was the origin of the anomalous shape of the bipolar bands. Actually, it has been reported that the anomalous IR properties including surface enhanced IR absorption, Fano-like IR effects, and abnormal IR effects, are strongly correlated to the structure and size of nanomaterials. We have observed the transformation among the three anomalous IR properties from experimental results^{30,37} and provided reasonable theoretical explanations^{38,39}. The strong interaction between the nanoparticles, and the collective interaction between CO adsorbates and the nanoparticles may be the origins of the anomalous IR properties of low-dimensional nanomaterials. It is evident to understand the origins of the anomalous IR properties of low-dimensional nanomaterials, further studies involving not only experimental approaches but also theoretical analysis would be worthwhile.

4 Conclusions

In summary, well monodispersed PtNi nanoparticles were synthesized by the galvanic displacement reaction and chemical reduction. The PtNi alloy nanoparticles are spherical and with an average diameter of about 52.2 nm, which have been approved by the results of TEM. Cyclic voltammetric results demonstrate that the PtNi nanomaterials display a better catalytic property than a bulk Pt for CO oxidation. It has been measured that the current peak potential of CO_{ad} oxidation on PtNi/GC electrode is shifted negatively by 75 mV in comparison with the value measured on a bulk Pt electrode. *In situ* electrochemical FTIRs employing CO adsorption as probe reaction demonstrate that the PtNi nanomaterials exhibit characteristics of bipolar IR feature of the CO_{ad} bands with an enhancement factor (A_{IR}) as high as 73.6. The intensity of the IR absorption of CO adsorbed on PtNi nanoparticles is so strong that we can observe and study the asymmetrical bipolar IR spectrum in the *in situ* single beam spectra clearly. The substrate materials do not affect significantly the anomalous IR features, as illustrated by the similar anomalous IR features observed for CO adsorbed on both PtNi/GC and PtNi/Au electrodes. These results described in this paper are of importance in controlled-synthesis of nanomaterials and understanding the fundamental of anomalous IR properties of low-dimensional nanomaterials.

References

- (1) Cheng, A.; Holt-Hindle, P. *Chem. Rev.* **2010**, *110*, 3767. doi: 10.1021/cr9003902
- (2) Ghosh, T. B.; Leonard, M.; Zhou, Q.; Disalvo, F. *Chem. Mater.* **2010**, *22*, 2190. doi:10.1021/cm9018474
- (3) Liu, F.; Lee, J. Y.; Zhou, W. J. *Small* **2006**, *2*, 121
- (4) Liu, F.; Lee, J. Y.; Zhou, W. J. *J. Phys. Chem. B* **2004**, *108*, 17959. doi:10.1021/jp0472360
- (5) Jayasayee, K.; Van Veen, J. A. R.; E. Hensen, J. M.; de Bruijn, F. A. *Electrochim. Acta* **2011**, *56*, 7235. doi:10.1016/j.electacta.2011.03.043
- (6) Hasche, F.; Oezaslan, M.; Strasser, P. *J. Phys. Chem. C* **2012**, *116*, 159, 25
- (7) Deivaraj, T. C.; Chen, W.; Lee, J. Y. *J. Mater. Chem.* **2003**, *13*, 2555. doi:10.1039/b307040a
- (8) Mandal, M.; Kundu, S.; Sau, T. K.; Yusuf, S. M.; Pal, T. *Chem. Mater.* **2003**, *15*, 3710. doi:10.1021/cm030246d
- (9) Ahrenstorf, K.; Albrecht, O.; Heller, H.; Kornowski, A.; Görlitz, D.; Weller, H. *Small* **2007**, *3*, 271
- (10) Cheng, F.; Ma, H.; Chen, J. *Inorg. Chem.* **2007**, *46*, 788. doi: 10.1021/ic061712e
- (11) Sun, Y.; Wiley, B.; Li, Z.; Xia, Y. *J. Am. Chem. Soc.* **2004**, *126*, 9399. doi:10.1021/ja048789r
- (12) Liang, H. P.; Zhang, H. M.; Hu, J. S.; Guo, Y. G.; Wan, L. J.; Bai, C. L. *Angew. Chem. Int. Edit.* **2004**, *43*, 1540.
- (13) Zhou, X. W.; Chen, Q. S.; Zhou, Z. Y.; Sun, S. G. *J. Nanosci. Nanotech.* **2009**, *9*, 2392. doi:10.1166/jnn.2009.SE34
- (14) Park, J. I.; Cheon, J. *J. Am. Chem. Soc.* **2001**, *123*, 5743. doi: 10.1021/ja0156340
- (15) Vasquez, Y.; Sra, A. K.; Schaak, R. E. *J. Am. Chem. Soc.* **2005**, *127*, 12504. doi:10.1021/ja054442s
- (16) Zeng, J.; Huang, J.; Lu, W.; Wang, X.; Wang, B.; Zhang, S.; Hou, J. *Adv. Mater.* **2007**, *19*, 2172
- (17) Zhou, X. W.; Zhang, R. H.; Zhou, Z. Y.; Sun, S. G. *J. Power Sources* **2011**, *196*, 5844. doi:10.1016/j.jpowsour.2011.02.088
- (18) Wang, J. X.; Ma, C.; Choi, Y. M.; Su, D.; Zhu, Y. M.; Liu, P.; Si, R.; Vukmirovic, M. B.; Zhang, Y.; Adzic, R. R. *J. Am. Chem. Soc.* **2011**, *133*, 13551. doi:10.1021/ja204518x
- (19) Zhou, X. W.; Gan, Y. L.; Sun, S. G. *Acta Phys. -Chim. Sin.* **2012**, *28*, 2071. [周新文, 甘亚利, 孙世刚. 物理化学学报, **2012**, *28*, 2071.] doi: 10.3866/PKU.WHXB201205031
- (20) Jiang, Y. X.; Sun, S. G.; Ding, N. *Chem. Phys. Lett.* **2001**, *344*, 463. doi:10.1016/S0009-2614(01)00812-0
- (21) Fano, U. *Phys. Rev.* **1961**, *124*, 1866. doi:10.1103/PhysRev.124.1866
- (22) Krauth, O.; Fahsold, G.; Pucci-Lehmann, A. *J. Mol. Struct.* **1999**, *483*, 237
- (23) Wang, H. C.; Sun, S. G.; Yan, J. W.; Yang, H. Z.; Zhou, Z. Y. *J. Phys. Chem. B* **2005**, *109*, 4309. doi:10.1021/jp046313o
- (24) Chen, Q. S.; Sun, S. G.; Yan, J. W.; Li, J. T.; Zhou, Z. Y. *Langmuir* **2006**, *22*, 10575. doi:10.1021/la0615037
- (25) Zhou, X. W.; Zhang, R. H.; Zeng, D. M.; Sun, S. G. *J. Solid State Chem.* **2010**, *183*, 1340. doi:10.1016/j.jssc.2010.04.003
- (26) Gong, H.; Sun, S. G.; Li, J. T.; Chen, Y. J.; Chen, S. P. *Electrochim. Acta* **2003**, *48*, 2933. doi:10.1016/S0013-4686(03)

- 00358-X
- (27) Lin, W. F.; Sun, S. G. *Electrochim. Acta* **1996**, *41*, 803. doi: 10.1016/0013-4686(95)00331-2
- (28) Priebe, A.; Fahsold, G.; Pucci, A. *Surf. Sci.* **2001**, *90*, 482
- (29) Zhu, Y.; Uchida, H.; Watanabe, M. *Langmuir* **1999**, *15*, 8757. doi:10.1021/la990835r
- (30) Gong, H.; Sun, S. G.; Li, J. T.; Chen, Y. J.; Chen, S. P. *Electrochim. Acta* **2003**, *48*, 2933. doi:10.1016/S0013-4686(03)00358-X
- (31) Lin, Y.; Sun, S. G. *Electrochim. Acta* **1998**, *44*, 1153. doi: 10.1016/S0013-4686(98)00218-7
- (32) Russell, J. W.; Severson, M.; Scanlon, K.; Overend, J.; Bewick, A. *J. Phys. Chem.* **1983**, *87*, 293. doi:10.1021/j100225a024
- (33) Kunitatsu, K.; Seki, H.; Golden, W. G.; Golden, J. G.; Philpott, M. R. *Surf. Sci.* **1985**, *158*, 596. doi:10.1016/0039-6028(85)90332-2
- (34) Pecharrómán, C.; Cuesta, A.; Gutiérrez, C. *J. Electroanal. Chem.* **2004**, *563*, 91. doi:10.1016/j.jelechem.2003.09.013
- (35) Christensen, P. A.; Hamnett, A.; Munk, J.; Troughton, G. L. *J. Electroanal. Chem.* **1994**, *370*, 251. doi:10.1016/0022-0728(93)03168-O
- (36) Ortiz, R.; Cuesta, A.; Márquez, O. P.; Márquez, J. M.; Méndez, J. A.; Gutiérrez, C. *J. Electroanal. Chem.* **1999**, *465*, 234. doi: 10.1016/S0022-0728(99)00099-6
- (37) Bjerke, A. E.; Griffiths, P. R.; Theiss, W. *Anal. Chem.* **1999**, *71*, 1967. doi:10.1021/ac981093u
- (38) Wu, C. X.; Lin, H.; Chen, Y. J.; Li, W. X.; Sun, S. G. *J. Chem. Phys.* **2004**, *121*, 1553. doi:10.1063/1.1763135
- (39) Su, Z. F.; Sun, S. G.; Wu, C. X.; Cai, Z. P. *J. Chem. Phys.* **2008**, *129*, 044707. doi:10.1063/1.2953441